CHALCOGENIDE-CARBON NANOFIBER AND PREPARATION METHOD THEREFOR

PATENT NUMBER 20180223455	DOCUMENT ID US 20180223455 A1		DATE PUBLISHEE 2018-08-09)	
INVENTOR INFORMATION					
NAME	CIT	Y	STATE	ZIP CODE	COUNTRY
JOO; Young Chang NAM; Dae Hyun	Seo Seo	-	N/A N/A	N/A N/A	KR KR
APPLICATION NO 15/749691		TE FILEC 6-07-29)		
FOREIGN APPLICATION PRIORITY DATA					
COUNTRY KR	APPLICA 10-2015-01				ON DATE
US CLASS CURREN	T:				
1/1					
CPC CURRENT					
TYPE	СР	С		DATE	
CPCI		1 F 9/10		2013-01-01	
CPCI		1 D 5/0038		2013-01-01	
CPCI CPCI		1 F 9/14 1 D 5/0046		2013-01-01 2013-01-01	
CPCI		1 F 1/10		2013-01-01	
CPCI		1 F 11/06		2013-01-01	
CPCA	D 0	1 F 9/22		2013-01-01	

CPCA CPCA CPCA CPCA

KWIC Hits

CPC ORIGINAL CLASSIFICATION GROUP

D D01F D01F11/06 20130101 F I 20180809 US D D10B D10B2101/10 20130101 L A 20180809 US D D01F D01F9/10 20130101 L I 20180809 US D D01D D01D5/0046 20130101 L I 20180809 US

INTERNATIONAL PUBLICATION CLASSIFICATION GROUP

D 10 B 2101/10

Y 10 T 428/30

B 32 B 9/007

2013-01-01

2015-01-15

2013-01-01

20060101 A D01F D01F11/06 F I B US H 20180809 20060101 A D01D D01D5/00 L I B US H 20180809 20060101 A D01F D01F9/10 L I B US H 20180809

PUBLICATION KIND CODE

CURRENT CPC CLASSIFICATION GROUP

D D01F D01F9/10 20130101 L I B H 20180809 US D D01F D01F9/22 20130101 L A B H 20191011 US D D10B D10B2101/10 20130101 L A B H 20180809 US D D01D D01D5/0038 20130101 L I B H 20191011 US D D01F D01F9/14 20130101 L I B H 20191011 US Y Y10T Y10T428/30 20150115 L A B H 20210402 US D D01D D01D5/0046 20130101 L I B H 20180809 US B B32B B32B9/007 20130101 L A B H 20210402 US D D01F D01F1/10 20130101 L I B H 20191011 US D D01F D01F1/10 20130101 L I B H 20191011 US

Abstract

In order to provide a method for preparing a chalcogenide-carbon nanofiber, capable of implementing oxidation resistance characteristics and process simplification, the present invention provides a method for preparing a chalcogenide-carbon nanofiber and a chalcogenide-carbon nanofiber implemented by using the same, the method comprising the steps of: forming a chalcogenide precursor-organic nanofiber comprising a chalcogenide precursor and an organic material; and forming a chalcogenide-carbon nanofiber by selectively and oxidatively heat treating the chalcogenide precursor-organic nanofiber such that the carbon of the organic material is oxidized and the chalcogenide is reduced at the same time, wherein the oxidation reactivity of the chalcogenide is lower than that of carbon, the selective and oxidative heat treatment is carried out through one heat treatment step instead of a plurality of heat treatment steps, and the chalcogenide can form a chalcogenide-carbon nanofiber having a structure formed with at least one layer according to an oxygen partial pressure at which the selective and oxidative heat treatment treatment is carried out.

Background/Summary

TECHNICAL FIELD

[0001] The present invention relates to a nanofiber and a manufacturing method thereof, and more particularly, to a chalcogenide-carbon nanofiber and a manufacturing method of the same.

BACKGROUND ART

[0002] Recently, there is a growing demand for a new nano processing technology or nanomaterials in various fields due to the development of nano-related technology. While electrospinning, split microfiber stretching, melt blowing, and the like are among the methods for producing nanofibers, electrospinning is the most useful manufacturing method of them. This is because it is relatively easy to process compared to other production methods, has less limitation in the use of materials, and can produce macro-sized shapes having pores of nano units.

[0003] However, it is difficult to control because of the irregularity of formation of nanofibers and nonwoven fabrics by electrospinning. Therefore, it is difficult to produce special shapes or patterns based on nanofibers through electrospinning, and it is difficult to find research and equipment related to the production of fine features.

DISCLOSURE OF THE INVENTION

Technical Problem

[0004] The object of the present invention is to solve various problems including the above problems, and includes a method of simultaneously forming a two (2)-dimensional structure of a nanofiber and a composite material, and various structures manufactured thereby. Particularly, according to the disclosure, after fabrication of nanofibers by electrospinning, the process is

systematized through variable control of oxygen partial pressure in a subsequent heat treatment, and the structure and properties of chalcogenide-carbon nanofibers fabricated according to each condition are presented. In addition, it was impossible to control the structure because the role of ambient was limited to reducing the precursor in a hydrogen or nitrogen gas atmosphere to make chalcogenide nanofibers. The bandgap of chalcogenides in a plate shape varies with the number of layers and the structure and, therefore, it is possible to control various characteristics such as electrical characteristics, catalytic characteristics, and optical characteristics. Conventionally, there was no technology to control such structure in the form of carbon nanofibers. The subsequent heat treatment process proposed in the present disclosure is a technique that overcomes the existing limitations because the structure of such chalcogenides can be customized and controlled through process variables. Furthermore, since it proceeds based on gas-phase reactions, the production efficiency in process is also excellent.

[0005] The present invention has been made to solve the above-mentioned problems. However, these problems are for illustrative purposes only and do not limit the scope of the present invention.

Technical Solution

[0006] According to one aspect of the present invention, a method of for manufacturing a chalcogenide-carbon nanofiber is provided. The method includes: forming a chalcogenide precursor-organic nanofiber including a chalcogenide precursor and an organic material; and forming a chalcogenide-carbon nanofiber through selective oxidation heat treatment of the chalcogenide precursor-organic nanofiber such that carbon of the organic material is oxidized and, simultaneously, the chalcogenide is reduced, wherein the chalcogenide has lower oxidation reactivity than carbon, wherein the selective oxidation heat treatment is performed in one heat treatment step rather than a plurality of heat treatment steps, and wherein the chalcogenide-carbon nanofiber is formed to have a structure in which chalcogenides are stacked in at least one layer depending on the magnitude of oxygen partial pressure under which the selective oxidation heat treatment is performed.

[0007] In the method of manufacturing a chalcogenide-carbon nanofiber, by controlling the decomposition degree of the carbon through the oxygen partial pressure at which the selective oxidation heat treatment is performed, the number of layers and the length of the chalcogenide in a plate shape may be controlled.

[0008] In the method of manufacturing a chalcogenide-carbon nanofiber, wherein the selective oxidation heat treatment may be performed in an atmosphere of a first oxygen partial pressure to a second oxygen partial pressure, the first oxygen partial pressure having a value relatively lower than the second oxygen partial pressure, wherein, when the chalcogenide precursor-organic nanofiber is heat-treated in an atmosphere having an oxygen partial pressure lower than the first oxygen partial pressure, the chalcogenide of the chalcogenide precursor may be reduced and the carbon of the organic material is also reduced by pyrolysis, wherein, when the chalcogenide precursor higher than the second oxygen partial pressure, the chalcogenide of the chalcogenide of the chalcogenide precursor may be reduced and the carbon of the second oxygen partial pressure, the chalcogenide of the chalcogenide of the chalcogenide precursor may be reduced and the carbon of the second oxygen partial pressure, the chalcogenide of the chalcogenide of the chalcogenide precursor may be reduced and the carbon of the second oxygen partial pressure, the chalcogenide of the chalcogenide precursor may be oxidized and the carbon of the organic material may also be oxidized by combustion.

[0009] In the method of manufacturing a chalcogenide-carbon nanofiber, wherein, when the chalcogenide precursor-organic nanofiber is heat-treated in an atmosphere of the first oxygen partial pressure to the second oxygen partial pressure, residual carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber is oxidized may be able to support the structure of the chalcogenide-organic nanofiber, and wherein, when the chalcogenide precursor-organic nanofiber is heat-treated in an atmosphere of having an oxygen partial pressure higher than the second oxygen partial pressure, residual carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber which remain after the second oxygen partial pressure, residual carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber is oxidized may not be able to support the structure of the chalcogenide-carbon nanofiber

[0010] In the method of manufacturing a chalcogenide-carbon nanofiber, wherein, when the chalcogenide precursor-organic nanofiber is heat-treated in an atmosphere of the first oxygen partial pressure to the second oxygen partial pressure, the chalcogenide in a plurality of layers is formed via controlled reaction kinetics depending on the concentration gradient of residual carbon

which remain after carbon of the organic material is oxidized such that the chalcogenide may have a structure of a plurality of layers distributively arranged in the chalcogenide-carbon nanofiber.

[0011] In the method of manufacturing a chalcogenide-carbon nanofiber, wherein, when the selective oxidation heat treatment is performed in an atmosphere of the second oxygen partial pressure, the chalcogenide may have a structure of more layers compared with when the selective oxidation heat treatment is performed in an atmosphere of the first oxygen partial pressure.

[0012] In the method of manufacturing a chalcogenide-carbon nanofiber, wherein the chalcogenide precursor may include ammonium tetrathiomolybdate (ATTM), and wherein the organic material may include polyacrylonitrile (PAN) that forms a hydrogen bond with the ammonium tetrathiomolybdate.

[0013] The method of manufacturing a chalcogenide-carbon nanofiber, may further comprise: forming a chalcogenide-carbon nanofiber through a selective oxidation heat treatment of the chalcogenide precursor-organic nanofiber such that carbon of the organic material is oxidized and, simultaneously, the chalcogenide precursor is reduced to a chalcogenide.

[0014] In the method of manufacturing a chalcogenide-carbon nanofiber, wherein the step of forming a chalcogenide-carbon nanofiber through selective oxidation heat treatment of the chalcogenide precursor-organic nanofiber may include: decomposing some of the carbon constituting the chalcogenide precursor-organic nanofiber by combustion rather than pyrolysis.

[0015] In the method of manufacturing a chalcogenide-carbon nanofiber, wherein the chalcogenide may include molybdenum disulfide (MoS.sub.2) or tungsten disulfide (WS.sub.2).

[0016] In the method of manufacturing a chalcogenide-carbon nanofiber, wherein the chalcogenide may include a compound of a transition metal and selenium (Se).

[0017] According to another aspect of the present invention, there is provided a method of manufacturing a chalcogenide-carbon nanofiber. The method comprises: preparing a solution containing ammonium tetrathiomolybdate (ATTM), polyacrylonitrile (PAN) and dimethyl formamide (DMF), forming a molybdenum disulfide-organic nanofiber by inducing a electrostatic repulsion by applying a voltage to the solution, and forming a molybdenum disulfide-carbon nanofiber through a selective oxidation heat treatment of the molybdenum disulfide-organic nanofiber such that carbon in the polyacrylonitrile is oxidized and, simultaneously, the molybdenum disulfide (MoS.sub.2) is reduced, wherein the selective oxidation heat treatment steps, and wherein the molybdenum disulfide-carbon nanofiber is formed to have a structure laminated with at least one layer depending on the magnitude of oxygen partial pressure under which the selective oxidation heat treatment is performed.

[0018] According to yet another aspect of the present invention, there is provided a chalcogenidecarbon nanofiber. The chalcogenide-carbon nanofibers are fabricated by the above-described method.

[0019] According to yet another aspect of the present invention, there is provided a chalcogenidecarbon nanofiber. The chalcogenide-carbon nanofiber may be such that a chalcogenide having a structure formed by at least one layer is distributively arranged in a structure in which carbon is aggregated.

[0020] The chalcogenide connected in series may be laminated with at least two layers.

[0021] The chalcogenide precursor-organic nanofiber comprising the chalcogenide precursor and the organic material may be fabricated through a selective oxidation heat treatment in one heat treatment step in an atmosphere of first oxygen partial pressure to a second oxygen partial pressure in which carbon of the organic material is oxidized and, simultaneously, the chalcogenide is reduced.

[0022] The chalcogenide may be subject to a selective oxidation heat treatment in an atmosphere of a third oxygen partial pressure higher than the first oxygen partial pressure and lower than the

second oxygen partial pressure and may be in the form of a single layer having a certain length and dispersed in the interior of a structure where the carbon is aggregated.

[0023] The chalcogenide may be subject to a selective oxidation heat treatment in an atmosphere of a fourth oxygen partial pressure higher than the third oxygen partial pressure and lower than the second oxygen partial pressure and may be in the form of a multi-layer having a certain length and dispersed in the interior of a structure where the carbon is aggregated.

[0024] By controlling the decomposition degree of carbon through the oxygen partial pressure in which the selective oxidation heat treatment is performed, the number of layers and the length of the chalcogenide in a plate shape may be controlled.

Advantageous Effects

[0025] According to one embodiment of the present invention as described above, it is possible to provide a method of manufacturing a chalcogenide-carbon nanofiber capable of realizing oxidation resistance characteristics and process simplification. The 2-dimensional structure and the composite material for improving the functionality of the nanofiber can be simultaneously realized through variable control in the process. Various performances can be realized according to the structure of the chalcogenide-carbon nanofibers fabricated by the disclosed process, and thus it can be applied to various fields. Of course, the scope of the present invention is not limited by these effects.

Description

BRIEF DESCRIPTION OF THE DRAWINGS

[0026] FIG. 1 is a flow chart illustrating a method of manufacturing a chalcogenide-carbon nanofiber according to an embodiment of the present invention.

[0027] FIG. 2 is a diagram illustrating a step of forming a chalcogenide precursor-organic nanofiber by electrospinning in a method of manufacturing a chalcogenide-carbon nanofiber according to an embodiment of the present invention.

[0028] FIGS. 3 to 6 are diagrams illustrating the concept of a selective oxidation heat treatment process in the method of manufacturing a chalcogenide-carbon nanofiber according to an embodiment of the present invention.

[0029] FIG. 7 shows photographs of chalcogenide-carbon nanofibers fabricated by a method of manufacturing a chalcogenide-carbon nanofiber according to a comparative example and some embodiments of the present invention.

[0030] FIGS. 8 and 9 are graphs schematically showing the structure and hydrogen evolution reaction (HER) performance of the chalcogenide-carbon nanofibers fabricated by the method of manufacturing chalcogenide-carbon nanofibers according to the comparative example and some embodiments of the present invention.

[0031] FIG. 10 shows photographs of a tungsten disulfide-carbon nanofiber fabricated by a method of manufacturing a chalcogenide-carbon nanofiber according to a comparative example of the present invention.

[0032] FIG. 11 shows photographs of and graphs evaluating hydrogen evolution reaction performance of a tungsten disulfide-carbon nanofiber fabricated by a method of manufacturing a chalcogenide-carbon nanofiber according to embodiments of the present invention.

[0033] FIG. 12 shows a result of charging and discharging tests when a chalcogenide-carbon nanofiber fabricated by a method of manufacturing a chalcogenide-carbon nanofiber according to an embodiment of the present invention was used as an anode material for a lithium ion battery.

EXPLANATION OF REFERENCE NUMERALS

[0034] **10**: syringe

[0035] **11**: tip

[0036] 12: high voltage source

[0037] 14: collector

[0038] 22: solution

[0039] 24: chalcogenide precursor-organic nanofiber

[0040] 24-1: chalcogenides-carbon nanofiber

[0041] **24***a*: transition metal (molybdenum)

[0042] 24a-1: oxidized transition metal

[0043] 24b: carbon

[0044] 24c: sulfur

[0045] 24d: nitrogen

[0046] 24e: chalcogenide (molybdenum disulfide)

[0047] 30: oxygen

[0048] 34: carbon monoxide

[0049] 35: carbon dioxide

MODE OF THE INVENTION

[0050] Hereinafter, embodiments of the present invention will be described in detail with reference to the accompanying drawings. It will be apparent to those skilled in the art that the present invention is not limited to the disclosed embodiments, but may be embodied in many different forms and should not be construed as limited to the embodiments set forth herein. The following embodiments are provided to fully inform those skilled in the art of the category of the present invention. Also, for convenience of explanation, the components may be exaggerated or reduced in size.

[0051] FIG. 1 is a flow chart illustrating a method of manufacturing a chalcogenide-carbon nanofiber according to an embodiment of the present invention.

[0052] Referring to FIG. 1, a method of manufacturing a chalcogenide-carbon nanofiber according to an embodiment of the present invention includes the steps of: providing a solution containing a chalcogenide precursor, an organic material, and a solvent (S10); forming a chalcogenide precursor-organic nanofiber from the solution by electrospinning using an electrostatic repulsion generated by applying high voltage to the solution (S20); and forming a chalcogenide-carbon nanofiber through selective oxidation heat treatment of the chalcogenide precursor-organic nanofiber such that carbon of the organic material is oxidized and, simultaneously, the chalcogenide precursor is reduced to a chalcogenide. In particular, the selective oxidation heat treatment is performed in one heat treatment step rather than a plurality of heat treatment steps.

[0053] In addition, the selective oxidation heat treatment is performed in an atmosphere of a first oxygen partial pressure to a second oxygen partial pressure. Particularly, a chalcogenide-carbon nanofiber may be formed to have a structure in which chalcogenides are stacked in at least one layer depending on the magnitude of oxygen partial pressure under which the selective oxidation heat treatment is performed. Here, the criteria for the first oxygen partial pressure and the second oxygen partial pressure are as follows. The magnitude of the second oxygen partial pressure is larger than the magnitude of the first oxygen partial pressure.

[0054] Therefore, when the chalcogenide precursor-organic nanofibers are heat-treated in an atmosphere having an oxygen partial pressure lower than the first oxygen partial pressure, the chalcogenide of the chalcogenide precursor is reduced and the carbon of the organic material is also reduced. On the other hand, when the chalcogenide precursor-organic nanofibers are heat-treated in an atmosphere having an oxygen partial pressure higher than the second oxygen partial pressure, the chalcogenide of the chalcogenide precursor is oxidized and the carbon of the organic substance is also oxidized.

[0055] When the chalcogenide precursor-organic nanofibers are heat-treated in an atmosphere of the first oxygen partial pressure to the second oxygen partial pressure, residual carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber is oxidized can support the structure of the chalcogenide-organic nanofiber. When the chalcogenide precursor-organic nanofiber is heat-treated in an atmosphere having an oxygen partial pressure higher than the second oxygen partial pressure, however, residual carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide-carbon nanofiber.

[0056] Furthermore, by controlling the decomposition degree of carbon through the oxygen partial pressure in the section in which the selective oxidation heat treatment is performed, the structure such as the number of layers and the length of the chalcogenide in a plate shape can be controlled. Particularly, the chalcogenide has various bandgaps according to the number of layers and the length and, therefore, it is possible to control various characteristics such as electrical characteristics, catalyst characteristics, and optical characteristics. Since the selective oxidation heat treatment can control the structure of the chalcogenide through the process parameters in a customized manner, it is possible to overcome the limitations of the conventional technologies, and it is advantageous in the production efficiency in the process because it proceeds based on a gas-phase reaction.

[0057] Meanwhile, the chalcogenide should have lower oxidation reactivity than carbon, and the chalcogenide may include a transition metal such as molybdenum or tungsten. These transition metals should have high reactivity with chalcogen such as sulfur (S) and selenium (Se). Hereinafter, for convenience of explanation, various embodiments will be described where the chalcogenide is molybdenum disulfide (MoS.sub.2). However, the technical idea of the present invention can be applied not only to molybdenum disulfide but also to compounds containing any transition metal having lower oxidation reactivity than carbon. For example, the technical idea of the present invention can be applied to a compound of transition metal and sulfur, a compound of transition metal and selenium, and the like.

[0058] FIG. 2 is a diagram illustrating a step of forming a chalcogenide precursor-organic nanofiber by electrospinning in a method of manufacturing a chalcogenide-carbon nanofiber according to an embodiment of the present invention.

[0059] Hereinafter, an electrospinning method according to an embodiment of the present invention will be described with reference to FIG. 2. A solution **22** made by mixing a precursor of chalcogenide, an organic material and a solvent is received in a syringe **10** for electrospinning. Electrospinning is a simple and highly efficient method of making chalcogenide precursor-organic nanofibers **24** using electrostatic repulsion by applying a high voltage to the solution **22**. The solution **22** that is used to manufacture the chalcogenide precursor-organic nanofibers **24** may be comprised of a chalcogenide solid solution such as a chalcogenide precursor, an organic matrix (organic material), and a solvent.

[0060] The chalcogenide solid solution is a material containing the ion of the chalcogenide nanofiber to be made. The combination between the functional group in the solid solution and the organic matrix is important. Thus, it may be desirable to select materials having the same or similar functional groups and make the dispersion of the chalcogenide solid solution uniform.

[0061] For example, the chalcogenide precursor may comprise ammonium tetrathiomolybdate (ATTM) to make molybdenum disulfide (MoS.sub.2)-organic nanofibers. The organic material, i.e. the organic matrix, sustains the structure of the nanofibers **24** that are formed first through electrospinning. For example, to make chalcogenide precursor-organic nanofibers, the organic

matrix may include polyacrylonitrile (PAN) that forms a hydrogen bond with the ammonium tetrathiomolybdate.

[0062] Furthermore, the solvent should be able to dissolve both the chalcogenide solid solution and the organic matrix. For example, to make the chalcogenide precursor-organic nanofibers, dimethyl formamide (DMF) can be used as the solvent because of the relatively high solubility of ammonium tetrathiomolybdate and polyacrylonitrile in dimethylformamide (DMF).

[0063] For example, the chalcogenide precursor-organic nanofibers **24** are fabricated through electrospinning, which produces fibers by applying electrostatic repulsion to the solution **22**. Chalcogenide-carbon nanofibers can be obtained by subjecting the chalcogenide precursor-organic nanofiber **24** formed by electrospinning to subsequent het treatment (calcination) of oxidation and reduction. In the electrospinning, it is possible to easily control the thickness of the nanofibers generated according to the magnitude of the voltage of several tens of kV applied to the solution **22**, and it is also possible to make the length longer than 100 µm. It is also advantageous that the transmittance and conductivity can be further improved through arrangement of the nanofibers. Such chalcogenide precursor-organic nanofibers **24** are important in that they can provide a solution to overcome the limitations of conventional nanowires.

[0064] The process of forming the chalcogenide precursor-organic nanofibers **24** is greatly influenced by the properties of the solution **22**. The morphology of the chalcogenide precursor-organic nanofibers **24** fabricated by electrospinning depends on the viscosity and the surface tension of the solution **22**, the concentration and the molecular weight of organic matter, and the conductivity of the solvent. Among these properties, the viscosity of the solution **22** may have the greatest influence. When the viscosity is very low or very high, a bead is formed in the chalcogenide precursor-organic nanofiber **24**, which is not suitable for a transparent electrode. Furthermore, in order to obtain the nanofiber shape suitable for the transparent electrode, the conditions should be optimized by adjusting the viscosity along with other properties of the solution.

[0065] Meanwhile, in the process of forming the chalcogenide precursor-organic nanofibers **24**, there are electrospinning process parameters and environmental parameters besides the solution properties. Environmental parameters include humidity and temperature. Since the optimal conditions for electrospinning are fixed, the atmosphere parameter can be controlled by creating an environment that can meet them.

[0066] Parameters that affect the chalcogenide precursor-organics nanofibers **24** more directly than environmental parameters are electrospinning process parameters. Electrospinning process parameters include the magnitude of the voltage applied by the high voltage source **12**, the distance between the tip **11** and the collector **14**, and the feeding rate of the solution **22**. The applied voltage is related to the electrostatic repulsive force which directly affects the formation of the chalcogenide precursor-organic nanofiber **24** in the solution **22**. The larger the applied voltage is, the smaller the diameter of the chalcogenide precursor-organic nanofiber **24** in the solution **22**. Therefore, by establishing such solution properties and process parameters, it is possible to form nanofibers that is so optimized as to be applied to active layers of field effect transistor (FET), photodetectors in OLED or solar cell, gas sensors, piezoelectric sensors, lithium ion batteries, sodium ion batteries and hydrogen release catalysts.

[0067] Meanwhile, various methods other than the above-described electrospinning are possible as a method of forming chalcogenide-carbon nanofibers. For example, chalcogenide-carbon nanofibers may be formed by a method of sulfurization in which a transition metal precursor is heat-treated by flowing sulfur or a selenium gas, a method of thermal decomposition of thiosalt, a vapor-phase reaction of a metal oxide and a chalcogenide or a vapor transport and recrystallization of a chalcogenide powder.

[0068] Next, the process of selective oxidation heat treatment of the chalcogenide precursororganic nanofibers **24** in order to realize the chalcogenide-carbon nanofibers will be described in detail with reference to FIGS. 3 to 12 as follows.

[0069] FIGS. 3 to 6 are diagrams illustrating the concept of a selective oxidation heat treatment process in the method of manufacturing chalcogenide-carbon nanofibers according to an

embodiment of the present invention.

[0070] Referring to FIGS. 3 and 4, the selective oxidation heat treatment process in the method for manufacturing the chalcogenide-carbon nanofibers according to the present invention is based on a gas-solid reaction. FIG. 3 is an enlarged view of a part of the chalcogenide precursor-organic nanofiber **24** and the chalcogenide precursor should be represented as a chalcogenide bonded to a transition metal. For convenience of explanation, however, it will be assumed that one transition metal element **24***a* is distributively arranged in an organic material. In the chalcogenide precursor-organic nanofiber **24**, the carbon atoms **24***b* constituting the organic material are aggregated, and the transition metal elements **24***a* are uniformly dispersed between the carbon atoms **24***b*. At this time, if the selective oxidation heat treatment process is performed, it can be categorized into three different types as shown in FIG. 4.

[0071] First, (a) of FIG. 4 schematically shows the reaction in the portion denoted by R1 in FIG. 3. The oxygen (O.sub.2) **30** molecules react with the carbon **24***b* to be oxidized in the form of carbon dioxide (CO.sub.2) **35** and decomposed from the chalcogenide precursor-organic nanofiber **24**, such that chalcogenide-carbon nanofibers having a two (2)-dimensional nanostructure can be formed. At this time, the carbon **24***b* is decomposed by pyrolysis rather than being oxidized and completely decomposed by combustion, so that a considerable amount of residual carbon remains. The nanofibers fabricated by this method have a structure in which the chalcogenide particles are densely dispersed between the amorphous carbon atoms.

[0072] (b) of FIG. 4 schematically shows the reaction in the portion denoted by R2 in FIG. 3. The carbon monoxide (CO) **34** reacts with the carbon **24***b* to be oxidized in the form of carbon dioxide **35** or the like and decomposed from the chalcogenide precursor-organic nanofiber **24**, such that chalcogenide-carbon nanofibers having a two (2)-dimensional nanostructure can be formed. At this time, since the carbon **24***b* is decomposed by pyrolysis as described above, a considerable amount of residual carbon remains.

[0073] Finally, (c) of FIG. 4 schematically shows the reaction in the portion labeled R3 in FIG. 3. The oxygen **30** molecule reacts with the transition metal **24***a* such that a chalcogenide-carbon nanofiber is formed in the form of an oxidized transition metal **24***a*-**1** or the like. The oxidation may include not only oxygen but also nitridation by nitrogen gas (N.sub.2) or ammonium (NH.sub.3).

[0074] An embodiment of the present invention provides a selective oxidation method in which transition metal **24***a* is reduced and only carbon **24***b* is oxidized and decomposed using the difference in oxidation reactivity between carbon and transition metal. This is significant because it has both advantages of heat treatment in air atmosphere and advantages of heat treatment through self-reduction.

[0075] In addition, in the method for manufacturing a chalcogenide-carbon nanofiber according to an embodiment of the present invention, the selective oxidation heat treatment process is performed in an atmosphere of a first oxygen partial pressure to a second oxygen partial pressure (that is, not lower than the first oxygen partial pressure but not higher than the second oxygen partial pressure). When the chalcogenide compound precursor-organic nanofiber **24** is heat-treated in an atmosphere having an oxygen partial pressure lower than the first oxygen partial pressure, the transition metal **24***a* of the chalcogenide precursor is reduced and the carbon **24***b* of the organic material is also reduced. When the chalcogenide precursor-organic nanofiber **24** is heattreated in an atmosphere having an oxygen partial pressure higher than the second oxygen partial pressure, the transition metal **24***a* of the chalcogenide precursor is oxidized and the carbon **24***b* of the organic material is also oxidized. For example, the first oxygen partial pressure may be an oxygen partial pressure corresponding to an oxidation point of the carbon **24***b*, and the second oxygen partial pressure may be an oxygen partial pressure corresponding to an oxidation point of the transition metal **24***a*.

[0076] When the chalcogenide precursor-organic nanofiber **24** is heat-treated in the atmosphere of the first oxygen partial pressure to the second oxygen partial pressure, the carbon **24***b* in the chalcogenide precursor-organic nanofiber **24** is oxidized and residual carbon **24***b* may support the structure of the chalcogenide-carbon nanofibers **24**. If, however, the chalcogenide precursor-organic nanofiber **24** is heat-treated in an atmosphere of oxygen partial pressure higher than the second oxygen partial pressure, all of the carbon **24***b* in the chalcogenide-organic nanofibers **24**.

are oxidized so that oxides are formed instead of the transition metal sulfides. Thus, the oxygen partial pressure setting is important.

[0077] Referring to FIGS. 5 and 6, if it is assumed for example that a molybdenum disulfide precursor is used, the chalcogenide precursor-organic nanofiber **24** according to an embodiment of the present invention may have a structure in which carbon atoms **24***b* are systematically aggregated with molybdenum (Mo) **24***a*, sulfur (S) **24***c*, and nitrogen (N) **24***d* elements uniformly dispersed and disposed between them.

[0078] When a selective oxidation heat treatment process is performed on the chalcogenide precursor-organic nanofiber **24** fabricated through electrospinning by the method of manufacturing a chalcogenide-carbon nanofiber according to an embodiment of the present invention, carbon **24***b* and molybdenum **24***a* have different oxidation reactivity and, therefore, the reaction **2** (reaction between oxygen **30** and carbon **24***b*) and the reaction **3** (reaction between molybdenum **24***a* and sulfur **24***c*) may be induced while the reaction **1** (reaction between oxygen **30** and molybdenum **24***a*) and the reaction **24***b* and sulfur **24***c*) may be suppressed.

[0079] That is, the chalcogenide precursor-organic nanofiber **24** is a composite nanofiber that includes molybdenum disulfide **24***e* constituting ammonium tetrathiomolybdate (ATTM), which is a chalcogenide precursor, and polyacrylonitrile (hereinafter, referred to as PAN), which is an organic material. In order to make the chalcogenide precursor-organic nanofiber **24** into chalcogenide-carbon nanofiber, a subsequent heat treatment (selective oxidation heat treatment) process is required.

[0080] In addition, the process for reducing molybdenum disulfide in the selective oxidation heat treatment process according to an embodiment of the present invention may include reactions represented by the following formulas 1 and 2.

(NH.sub.4).sub.2MoS4.fwdarw.2NH.sub.3+H.sub.2S+MoS.sub.3 (Formula 1)

MoS.sub.3.fwdarw.MoS.sub.2+S (Formula 2)

[0081] By these reactions, the hydrogen released from the ammonium ion contained in ATTM reacts with sulfur to generate hydrogen sulfide (H.sub.2S) and ammonia (NH.sub.3), and molybdenum trisulfide (MoS.sub.3) is produced. Pure sulfur molybdenum (MoS.sub.2) phase can be obtained as sulfur (S) is released from the molybdenum trisulfide (MoS.sub.3).

[0082] Hereinafter, it will be described from the thermodynamic point of view that a selective oxidation heat treatment process can be performed in the method of manufacturing chalcogenide-carbon nanofibers according to an embodiment of the present invention.

[0083] The reduction of molybdenum disulfide (MoS.sub.2) from ammonium tetrathiomolybdate (ATTM), which is a chalcogenide precursor used in the selective oxidation heat treatment, is due to the reduction of molybdenum disulfide (MoS.sub.2) by autoreduction during the selective oxidation heat treatment. Therefore, the reaction of molybdenum disulfide generated in autoreduction must be considered in order to find the oxygen partial pressure required in the heat treatment.

[0084] First, it is necessary to identify the Gibbs free energy for oxidation reactions of carbon and molybdenum and the Gibbs free energy for sulfidation reactions of carbon and molybdenum in the Ellingham diagram. The actual Ellingham diagram shows the Gibbs free energy in the reactions in which molybdenum and carbon are oxidized at a given temperature and pressure. In order to find the oxygen partial pressure for selective oxidation heat treatment, the Gibbs free energy for the oxidation reactions and the sulfidation reactions of molybdenum and carbon obtained through a series of processes can be shown in the Ellingham diagram and compared with the actual oxidation reaction of molybdenum disulfide.

[0085] Meanwhile, the cross section of the chalcogenide-carbon nanofiber implemented by the above-described mechanism can be formed as shown in FIG. 6. (a) to (d) of FIG. 6 show that the length of the chalcogenide, the layered morphology, the structure, and the interlayer distance are controlled depending on the oxygen partial pressure.

[0086] Referring first to (a) of FIG. 6, as a structure of the chalcogenide-carbon nanofibers **24-1** subjected to a selective oxidation heat treatment in the atmosphere of the relatively smallest oxygen partial pressure, the chalcogenide **24***e* may be in the form of a relatively short single layer and dispersed uniformly in the interior of the chalcogenide-carbon nanofibers **24-1**, i.e. the structure where the carbon atoms **24***b* are aggregated.

[0087] Referring to (b) of FIG. 6, as a structure of the chalcogenide-carbon nanofibers **24-1** subjected to a selective oxidation heat treatment in an atmosphere of a relatively larger oxygen partial pressure than that of (a) of FIG. 6, the chalcogenide **24***e* may be in the form of a single layer in which the chalcogenide **24***e* elongates longer than that of (a) of FIG. 6 and may be dispersed evenly.

[0088] Referring to (c) of FIG. 6, as a structure of the chalcogenide-carbon nanofibers **24-1** subjected to a selective oxidation heat treatment in an atmosphere of relatively larger oxygen partial pressure than that of (b) of FIG. 6, more than two linearly-connected chalcogenides **24***e* may be stacked and dispersed evenly.

[0089] Referring, lastly, to (d) of FIG. 6, as a structure of the chalcogenide-carbon nanofibers **24-1** subjected to a selective oxidation heat treatment in an atmosphere of a relatively larger oxygen partial pressure than that of (c) of FIG. 6, more layers than those shown in (c) of FIG. 6 may be stacked to have a structure substantially filling the inside of the carbon nanofibers.

[0090] Therefore, by controlling the atmosphere of the oxygen partial pressure in the subsequent heat treatment after manufacturing the nanofibers by electrospinning, various hybrid nanofibers can be mass-produced from the same raw material, and various structures of nanofibers and functional particles can be produced.

[0091] Experimental examples of a method according to the present invention is provided below to facilitate understanding of the present invention. However, the following experimental examples are provided to aid understanding of the present invention, and the present invention is not limited by the following experimental examples.

[0092] In the experimental examples of the present invention, a solution was prepared by mixing ATTM, PAN and DMF, and a high voltage was applied to the solution to form a molybdenum disulfide precursor-organic nanofiber. Thereafter, molybdenum disulfide-carbon nanofiber samples were prepared by performing a selective oxidation heat treatment process at about 800° C. for about 5 hours by adjusting oxygen partial pressures to 0.1 torr, 0.2 torr, 0.3 torr, and 0.4 torr, respectively.

[0093] On the other hand, for comparison, a sample of comparative example 1 was prepared by using the same materials as those used in the above-mentioned experimental examples but performing heat treatment in an atmosphere of argon/hydrogen instead of oxygen partial pressure.

[0094] The structures of the samples were analyzed by a scanning electron microscope and a transmission electron microscope, and the hydrogen evolution reaction performance was tested. The hydrogen evolution reaction performance test was carried out by using molybdenum disulfide and an amorphous carbon electrode and using graphite as a counter electrode. Furthermore, in order to compare the results of the hydrogen evolution reaction performance test, as comparative example 2 of the present invention, a hydrogen evolution reaction performance test was conducted using a commercially available platinum (Pt) catalyst.

[0095] In addition, as an experimental example according to another embodiment of the present invention, a tungsten disulfide compound-organic nanofiber was prepared by using a precursor of tungsten disulfide compound and electrospinning in the same manner as in the above experimental example. Thereafter, a selective oxidation heat treatment was performed in an atmosphere of oxygen partial pressure at 0.1 torr and 0.2 torr, respectively, to prepare samples of tungsten disulfide-carbon nanofibers according to an embodiment of the present invention.

[0096] For comparison, a sample of comparative example 3 was prepared by using the same materials as those used in the above experimental examples but performing heat treatment in an atmosphere of argon/hydrogen instead of oxygen partial pressure.

[0097] FIG. 7 shows photographs of chalcogenide-carbon nanofibers implemented by a method of manufacturing a chalcogenide-carbon nanofiber according to a comparative example and some embodiments of the present invention.

[0098] Particularly, (a) of FIG. 7 shows photographs of a chalcogenide-carbon nanofiber according to comparative example 1 of the present invention, which is formed through a selective oxidation heat treatment of a chalcogenide precursor-organic nanofiber in an argon/hydrogen atmosphere; (b) of FIG. 7 shows photographs of a chalcogenide-carbon nanofiber according to experimental example 1 of the present invention, which is formed through a selective oxidation heat treatment of a chalcogenide precursor-organic nanofiber in atmosphere of oxygen partial pressure of 0.1 torr, in an atmosphere ranging from a first oxygen partial pressure to a second oxygen partial pressure; (c) of FIG. 7 shows photographs of chalcogenide-carbon nanofiber according to experimental example 2 of the present invention, which is formed through a selective oxidation heat treatment of a chalcogenide precursor-organic nanofiber in an atmosphere of oxygen partial pressure atmosphere of 0.2 torr, in an atmosphere ranging from a first oxygen partial pressure to a second oxygen partial pressure; (d) of FIG. 7 shows photographs of a chalcogenide-carbon nanofiber according to experimental example 3 of the present invention, which is formed through a selective oxidation heat treatment of chalcogenide precursor-organic nanofiber in an atmosphere of oxygen partial pressure atmosphere of 0.3 torr, in an atmosphere ranging from a first oxygen partial pressure to a second oxygen partial pressure; and (e) of FIG. 7 shows photographs of a chalcogenide-carbon nanofiber according to experimental example 4 of the present invention, which is formed through a selective oxidation heat treatment of a chalcogenide precursor-organic nanofiber in an atmosphere of oxygen partial pressure of 0.4 torr, in an atmosphere ranging from a first oxygen partial pressure to a second oxygen partial pressure. Since the structure of these nanofibers has been described above, the explanation thereof will be omitted.

[0099] FIGS. 8 and 9 are graphs schematically showing the structure and hydrogen evolution reaction performance of the chalcogenide-carbon nanofibers implemented by the method of manufacturing chalcogenide-carbon nanofibers according to the comparative example and some embodiments of the present invention.

[0100] First, (a) of FIG. 8 shows the results of the hydrogen evolution reaction performance test performed five times on the sample according to comparative example 1 of the present invention in which molybdenum disulfide-carbon nanofibers were produced in an argon/hydrogen atmosphere as described above with reference to (a) of FIG. 7. (b) of FIG. 8 shows the results of the hydrogen evolution reaction performance test performed five times on the sample according to experimental example 1 of the present invention, on which a selective oxidation heat treatment was performed in an atmosphere of oxygen partial pressure of 0.1 torr as described above with reference to (b) of FIG. 7. (c) of FIG. 8 shows the results of the hydrogen evolution reaction performance test performed five times on the sample according to experimental example 2 of the present invention, on which a selective oxidation heat treatment was performed five times on the sample according to experimental example 2 of the present invention, on which a selective oxidation heat treatment of 0.2 torr as described above with reference to (c) of FIG. 7. (d) of FIG. 8 shows the results of the hydrogen evolution reaction performed five times on the sample according to experimental example 3 of the present invention, on which a selective oxidation heat treatment was performed five times on the sample according to experimental example 3 of the present invention, on which a selective oxidation heat treatment was performed five times on the sample according to experimental example 3 of the present invention, on which a selective oxidation heat treatment was performed five times on the sample according to experimental example 3 of the present invention, on which a selective oxidation heat treatment was performed in an atmosphere of 0.4 torr as described above with reference to (e) of FIG. 7.

[0101] FIG. 9 shows graphs illustrating the results of hydrogen evolution reaction performance test using commercially available platinum (Pt) catalysts according to comparative example 2, in addition to the samples tested as shown in FIG. 8, and comparing the results with the samples tested as shown in FIG. 8.

[0102] Referring to FIGS. 8 and 9, the change in the current density with the change in the voltage drop as hydrogen is generated shows that the commercially available platinum catalyst of comparative example 2 of the present invention showed the best hydrogen evolution reaction performance. Furthermore, it can be seen that all the other samples have similar characteristics in hydrogen generation and slope although they have slightly lower performances.

[0103] FIG. 10 shows photographs of a tungsten disulfide-carbon nanofiber fabricated by a method of manufacturing a chalcogenide-carbon nanofiber according to comparative example 3 of the present invention, and FIG. 11 shows photographs of and graphs evaluating hydrogen evolution

reaction performance of a tungsten disulfide-carbon nanofiber fabricated by a method of manufacturing a chalcogenide-carbon nanofiber according to embodiments of the present invention.

[0104] Referring to FIG. 10, it can be seen that both a single layer and a multi-layer structure are observed in the structure of tungsten disulfide, which is a sample of comparative example 3 of the present invention.

[0105] On the other hand, FIG. 11 shows the samples fabricated by the embodiments of the present invention, i.e. the sample of experimental example 5 prepared in an atmosphere of oxygen partial pressure of 0.1 torr and the sample of experimental example 6 prepared in an atmosphere of oxygen partial pressure of 0.2 torr. The structures of these samples show that the tungsten disulfide is appropriately controlled. Furthermore, all of the samples according to the embodiments of the present invention were repeatedly tested for hydrogen evolution reaction performance, and as a result, show the change in the current density with the change in the voltage drop as hydrogen is generated.

[0106] FIG. 12 shows a result of charging and discharging test when a chalcogenide-carbon nanofiber fabricated by a method of manufacturing a chalcogenide-carbon nanofiber according to an embodiment of the present invention was used as an anode material for a lithium ion battery.

[0107] Referring to FIG. 12, when the molybdenum disulfide-carbon nanofiber according to experimental example 1 of the present invention is used as an anode material for a lithium ion battery, it has a capacity of 700 mAh/g or more and an electrical efficiency of about 98% or more during charging and discharging cycles. Therefore, the chalcogenide-carbon nanofiber according to an embodiment of the present invention is applicable to an anode material for a lithium ion battery.

[0108] As described above, the chalcogenide-carbon nanofibers fabricated by the method of manufacturing a chalcogenide-carbon nanofiber according to the embodiments of the present invention can be produced by controlling the atmosphere of the oxygen partial pressure in a heat treatment process following the electrospinning, thereby simultaneously controlling the structure and functional particles of the nanofibers. In addition, it is possible to produce various functional particles such as metals, metal oxides/nitrides/sulfides, and it is also possible to mass-produce various hybrid nanofibers from the same raw materials. Therefore, it is possible to develop a platform technology applicable to various industrial fields through the simplification of the manufacturing process.

[0109] While the present invention has been described with reference to embodiments shown in the drawings, various modifications and equivalents to the examples described herein will be readily apparent to those skilled in the art. Accordingly, the true scope of the present invention should be determined by the technical idea of the appended claims.

Claims

1. A method for manufacturing a chalcogenide-carbon nanofiber, the method comprising: forming a chalcogenide precursor-organic nanofiber including a chalcogenide precursor and an organic material; and forming a chalcogenide-carbon nanofiber through selective oxidation heat treatment of the chalcogenide precursor-organic nanofiber such that carbon of the organic material is oxidized and, simultaneously, the chalcogenide is reduced, wherein the chalcogenide has lower oxidation reactivity than carbon, wherein the selective oxidation heat treatment is performed in one heat treatment step rather than a plurality of heat treatment steps, and wherein the chalcogenide-carbon nanofiber is formed to have a structure in which chalcogenides are stacked in at least one layer depending on the magnitude of oxygen partial pressure under which the selective oxidation heat treatment is performed.

2. The method of claim 1, wherein, by controlling the decomposition degree of the carbon through the oxygen partial pressure at which the selective oxidation heat treatment is performed, the number of layers and the length of the chalcogenide in a plate shape is controlled.

3. The method of claim 1, wherein the selective oxidation heat treatment is performed in an atmosphere of a first oxygen partial pressure to a second oxygen partial pressure, the first oxygen

partial pressure having a value relatively lower than the second oxygen partial pressure, wherein, when the chalcogenide precursor-organic nanofiber is heat-treated in an atmosphere having an oxygen partial pressure lower than the first oxygen partial pressure, the chalcogenide of the chalcogenide precursor is reduced and the carbon of the organic material is also reduced, wherein, when the chalcogenide precursor-organic nanofiber is heat-treated in an atmosphere having an oxygen partial pressure higher than the second oxygen partial pressure, the chalcogenide of the chalcogenide precursor is oxidized and the carbon of the organic material is also oxidized.

4. The method of claim 3, wherein, when the chalcogenide precursor-organic nanofiber is heattreated in an atmosphere of the first oxygen partial pressure to the second oxygen partial pressure, residual carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber is oxidized can support the structure of the chalcogenide-organic nanofiber, and wherein, when the chalcogenide precursor-organic nanofiber is heat-treated in an atmosphere of having an oxygen partial pressure higher than the second oxygen partial pressure, residual carbon in the chalcogenide precursor-organic nanofiber which remain after the carbon in the chalcogenide precursor-organic nanofiber which the structure of the chalcogenide-carbon nanofiber.

5. The method of claim 3, wherein, when the chalcogenide precursor-organic nanofiber is heattreated in an atmosphere of the first oxygen partial pressure to the second oxygen partial pressure, the chalcogenide in a plurality of layers is formed via controlled reaction kinetics depending on the concentration gradient of residual carbon which remain after carbon of the organic material is oxidized such that the chalcogenide has a structure of a plurality of layers distributively arranged in the chalcogenide-carbon nanofiber.

6. The method of claim 3, wherein, when the selective oxidation heat treatment is performed in an atmosphere of the second oxygen partial pressure, the chalcogenide has a structure of more layers compared with when the selective oxidation heat treatment is performed in an atmosphere of the first oxygen partial pressure.

7. The method of claim 1, wherein the chalcogenide precursor includes ammonium tetrathiomolybdate (ATTM), and wherein the organic material includes polyacrylonitrile (PAN) that forms a hydrogen bond with the ammonium tetrathiomolybdate.

8. The method of claim 7, further comprising: forming a chalcogenide-carbon nanofiber through a selective oxidation heat treatment of the chalcogenide precursor-organic nanofiber such that carbon of the organic material is oxidized and, simultaneously, the chalcogenide precursor is reduced to a chalcogenide.

9. The method of claim 1, wherein the step of forming a chalcogenide-carbon nanofiber through selective oxidation heat treatment of the chalcogenide precursor-organic nanofiber includes: decomposing some of the carbon constituting the chalcogenide precursor-organic nanofiber by combustion rather than pyrolysis.

10. The method of claim 1, wherein the chalcogenide includes molybdenum disulfide (MoS.sub.2) or tungsten disulfide (WS.sub.2).

11. The method of claim 1, wherein the chalcogenide includes a compound of a transition metal and selenium (Se).

12. A method of manufacturing a chalcogenide-carbon nanofiber, the method comprising: preparing a solution containing ammonium tetrathiomolybdate (ATTM), polyacrylonitrile (PAN) and dimethyl formamide (DMF), forming a molybdenum disulfide-organic nanofiber by inducing a electrostatic repulsion by applying a voltage to the solution, and forming a molybdenum disulfide-carbon nanofiber through a selective oxidation heat treatment of the molybdenum disulfide-organic nanofiber such that carbon in the polyacrylonitrile is oxidized and, simultaneously, the molybdenum disulfide (MoS.sub.2) is reduced, wherein the selective oxidation heat treatment is performed in one heat treatment step rather than a plurality of heat treatment steps, and wherein the molybdenum disulfide-carbon nanofiber is formed to have a structure laminated with at least one layer depending on the magnitude of oxygen partial pressure under which the selective oxidation heat treatment is performed.

13. A chalcogenide-carbon nanofiber fabricated by a method of claim 1.

14. A chalcogenide-carbon nanofiber in which a chalcogenide having a structure formed by at least one layer is distributively arranged in a structure in which carbon is aggregated.

15. The chalcogenide-carbon nanofiber of claim 14, wherein the chalcogenide connected in series is laminated with at least two layers.

16. The chalcogenide-carbon nanofiber of claim 14, wherein the chalcogenide precursor-organic nanofiber comprising the chalcogenide precursor and the organic material is fabricated through a selective oxidation heat treatment in one heat treatment step in an atmosphere of first oxygen partial pressure to a second oxygen partial pressure in which carbon of the organic material is oxidized and, simultaneously, the chalcogenide is reduced.

17. The chalcogenide-carbon nanofiber of claim 16, wherein the chalcogenide is subject to a selective oxidation heat treatment in an atmosphere of a third oxygen partial pressure higher than the first oxygen partial pressure and lower than the second oxygen partial pressure and is in the form of a single layer having a certain length and dispersed in the interior of a structure where the carbon is aggregated.

18. The chalcogenide-carbon nanofiber of claim 17, wherein the chalcogenide is subject to a selective oxidation heat treatment in an atmosphere of a fourth oxygen partial pressure higher than the third oxygen partial pressure and lower than the second oxygen partial pressure and is in the form of a multi-layer having a certain length and dispersed in the interior of a structure where the carbon is aggregated.

19. The chalcogenide-carbon nanofiber of claim 14, wherein, by controlling the decomposition degree of carbon through the oxygen partial pressure in which the selective oxidation heat treatment is performed, the number of layers and the length of the chalcogenide in a plate shape are controlled.